

# Tetraphenylanthraquinone and Dihydroxybenzene-Tethered Conjugated Microporous Polymer for Enhanced CO<sub>2</sub> Uptake and Supercapacitive Energy Storage

Published as part of JACS Au special issue "Polymers for the Clean Energy Transition".

Mohamed Gamal Mohamed, Chia-Chi Chen, Mervat Ibrahim, Aya Osama Mousa, Mohamed Hammad Elsayed, Yunsheng Ye, and Shiao-Wei Kuo\*



Cite This: JACS Au 2024, 4, 3593–3605



Read Online

ACCESS |



Metrics & More



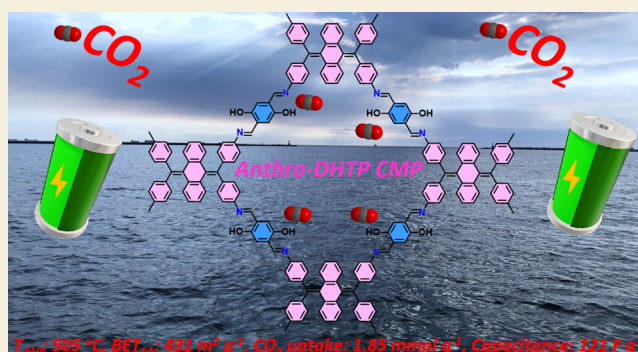
Article Recommendations



Supporting Information

**ABSTRACT:** Conjugated microporous polymers (CMPs) feature extended excellent porosity properties and fully conjugated electronic systems, making them highly effective for several uses, including photocatalysis, dye adsorption, CO<sub>2</sub> capture, supercapacitors, and so on. These polymers are known for their high specific surface area and adjustable porosity. To synthesize DHTP-CMPs (specifically TPE-DHTP CMP and Anthra-DHTP CMP) with abundant nitrogen (N) and oxygen (O) adsorption sites and spherical structures, we employed a straightforward Schiff-base [4 + 2] condensation reaction. This involved using 2,5-dihydroxyterephthalaldehyde (DHTP-2CHO) as the primary building block and phenolic OH group source, along with two distinct structures: 4,4',4'',4'''-(ethene-1,1,2,2-tetrayl)tetraaniline (TPE-4NH<sub>2</sub>) and 4,4',4'',4'''-(anthracene-9,10-diylidenebis(methanediylidene))tetraaniline (Anthra-4Ph-4NH<sub>2</sub>). The synthesized Anthra-DHTP CMP had a remarkable BET surface area (BET<sub>SA</sub>) of 431 m<sup>2</sup> g<sup>-1</sup>. Additionally, it exhibited outstanding thermal stability, as shown by a *T*<sub>d10</sub> of 505 °C. Furthermore, for practical implementation, the Anthra-DHTP CMP demonstrates a significant capacity for capturing CO<sub>2</sub>, measuring 1.85 mmol g<sup>-1</sup> at a temperature of 273 K and 1 bar. In a three-electrode test, the Anthra-DHTP CMP has a remarkable specific capacitance of 121 F g<sup>-1</sup> at 0.5 A g<sup>-1</sup>. Furthermore, even after undergoing 5000 cycles, it maintains a capacitance retention rate of 79%. Due to their outstanding pore characteristics, abundant N and O, and conjugation properties, this Anthra-DHTP CMP holds significant potential for CO<sub>2</sub> capture and supercapacitor applications. This work will pave the way for the development of materials based on DHTP-CMPs and their postmodification with additional groups, facilitating their use in photocatalysis, photodegradation, lithium battery applications, and so on.

**KEYWORDS:** tetraphenylanthraquinone, dihydroxybenzene, conjugated microporous polymers, CO<sub>2</sub> uptake, supercapacitors



## INTRODUCTION

The rapid increase in emissions of carbon dioxide (CO<sub>2</sub>) from the burning of fossil fuels presents significant challenges for global societies, including climate change, resource depletion, and elevated levels of environmental contamination.<sup>1–9</sup> Considerable endeavors are underway to mitigate CO<sub>2</sub> emissions employing developing affordable, environmentally friendly devices, aimed at fostering enduring advantages for our societies.<sup>10–13</sup> For instance, electrochemical energy storage (EES) systems like lithium-ion batteries (LIBs), supercapacitors (SCs), and water-splitting electrolyzers, are gaining prominence for energizing developing electronic devices due to their dependable functionality and flexibility.<sup>14–22</sup> LIBs and SCs are essential components of electronically stored energy systems. SCs offer numerous advantages over batteries, including extensive cycle longevity, good power densities,

and remarkable rate handling capacities, classifying them as large-capacity energy storage systems.<sup>22–27</sup> As a result, SCs are valuable for energy storage in industrial, transportation, and technological applications. Additionally, optimizing the power density and performance of supercapacitors necessitates careful consideration of the physical and chemical properties of the electrode materials. Therefore, to meet current demands, a shift toward advanced materials and products is required.<sup>28–30</sup>

**Received:** June 23, 2024

**Revised:** July 23, 2024

**Accepted:** August 9, 2024

**Published:** August 16, 2024



Supercapacitors may be classified into three primary categories according to their energy storage techniques: electric double-layer capacitors (EDLCs), battery-like materials, and pseudocapacitive devices.<sup>31</sup> SCs are widely utilized in various applications such as high-power supplies, multilevel inverters, advanced electric, and vehicles. Their extensive cycle life, high power density, excellent rate capability, wide operating temperature range, ultrafast charge/discharge rates, reversibility, and potential to meet increasing power demands make them a promising solution to address energy scarcity.<sup>32–36</sup> Despite progress, SCs still have lower energy densities compared to other rechargeable batteries, and their market share in energy storage devices remains relatively small.<sup>37,38</sup> To expand the applicability of SC devices, significant endeavors are being directed toward augmenting the energy densities of SCs and engineering materials endowed with multifaceted attributes, such as extensive surface areas, customizable porous architectures, highly conductive systems, and enhanced wettability.<sup>39</sup>

Conjugated polymers are often preferred in SCs due to their cost-effectiveness and widespread availability.<sup>40</sup> For example, electrodes made from polyaniline/composites nanostructures have shown a capacitance of 1221 F g<sup>-1</sup>.<sup>41</sup> Nonetheless, conventional conjugated polymers (for example, polypyrrole, polyaniline, and polythiophene) are typically devoid of porosity and lack sufficient durability.<sup>42</sup> Typically, their capacitance declines significantly after more than 1000 charge and discharge cycles. CMPs are a subset of porous organic polymers characterized by extended  $\pi$ -conjugated structures within their microporous networks, showing promising potential for use in supercapacitors.<sup>43,44</sup> These porous architectures, distinct from those found in linear polymers, are expected to significantly improve electron and ion transport capabilities.<sup>45</sup> CMPs are traditionally synthesized using various C–N or C–C coupling reactions, such as oxidative polymerization, Buchwald-Hartwig amination, Sonogashira-Hagihara coupling, Schiff base formation, Suzuki-Miyaura coupling, cyclotrimerization, Yamamoto coupling, and phenazine ring fusion methods.<sup>46–50</sup> These techniques have been utilized to fashion CMPs showcasing varied structural configurations and intrinsic attributes.<sup>51</sup> CMPs have been extensively studied for their applicability in various fields, including light-emitting diodes, energy conversion, chemosensing, energy storage, catalysis, and various subfields of biological sciences.<sup>52–57</sup> The suitability of CMPs as active electrode materials for supercapacitors has been assessed by studying CMPs with different structural variations.<sup>58–62</sup> Due to the diverse building blocks used in CMP structures, their ability to modify  $\pi$ -conjugated units, and their inherent tunability, CMPs often surpass traditional electrode materials in supercapacitor electrode design.<sup>63</sup> Tetraphenylanthraquinone (Anthra-4Ph) is a chemical molecule with a complex aromatic structure, featuring an anthraquinone core bonded to four phenyl groups. Renowned for its strong chemical stability and unique electrical properties, Anthra-4Ph has garnered interest across a wide range of scientific fields. Its conjugated structure allows for significant electron delocalization, making it a valuable component in organic electronics, photovoltaics, and photochemical applications.<sup>64,65</sup> Mohamed et al. synthesized An-CPOP-2 using Anthra-4Ph and 2,4,6-trichloro-1,3,5-triazine as monomers. The resulting material demonstrated a CO<sub>2</sub> capacity of 1.52 mmol g<sup>-1</sup> (6.7 wt %, 273 K) and capacitance of 98.4 F g<sup>-1</sup>.<sup>64</sup> Our group-prepared PET-Im CMP through

Sonogashira coupling with a specific capacitance of 63 F g<sup>-1</sup>.<sup>43</sup> According to Bhaumik and co-workers, TFR-NDA demonstrated a capacitance of 362 F g<sup>-1</sup> in a three-electrode setup.<sup>66</sup> The Pandey group disclosed that ITR-COF possesses a high specific capacitance of 182.6 F g<sup>-1</sup>, as measured at 0.3 A g<sup>-1</sup>.<sup>67</sup>

In this work, using a straightforward and effective Schiff-base condensation process, we successfully created two types of DHTP-CMPs with CH=N and phenolic units: TPE-DHTP CMP and Anthra-DHTP CMP. Anthra-DHTP CMP is notable for its exceptional thermal stability, with a thermal decomposition temperature ( $T_{d10}$ ) of 505 °C and a char yield of 68 wt %. Based on N<sub>2</sub> adsorption–desorption and CO<sub>2</sub> isotherms, Anthra-DHTP CMP has the largest BET<sub>SA</sub> of 431 m<sup>2</sup> g<sup>-1</sup> and the highest CO<sub>2</sub> absorption capacity (1.85 mmol g<sup>-1</sup> or 8.14 wt % at 273 K). Furthermore, Anthra-DHTP CMP demonstrated capacitance values of 121 F g<sup>-1</sup> in the GCD tests. This study showcases the efficient production of DHTP-CMPs through free-metal [4 + 2] condensation reaction and highlights their potential applications beyond supercapacitors and CO<sub>2</sub> absorption. These DHTP-CMP materials are also expected to show promise in related areas such as iodine and dye capture, photocatalysis, and other fields.

## EXPERIMENTAL SECTION

### Materials

4-Aminophenylboronic acid (BZB-NH<sub>2</sub>, 98%), tin (Sn, ≥99.8%), carbon tetrabromide (CBr<sub>4</sub>, 99%), tetrahydrofuran (THF), potassium carbonate (K<sub>2</sub>CO<sub>3</sub>, ≥99.8%), acetic acid (AcOH, ≥99%), anthraquinone (Anthra, 97%), toluene, triphenylphosphine (PPh<sub>3</sub>, 99%), 4,4'-diaminobenzophenone (BZP-2NH<sub>2</sub>, 97%), 1,4-dioxane (DO), mesitylene (98%), and acetone, tetrakis(triphenylphosphine)-palladium [Pd(PPh<sub>3</sub>)<sub>4</sub>, 98%], anhydrous magnesium sulfate (MgSO<sub>4</sub>, ≥99.5%), and sodium hydroxide (NaOH, ≥98%), 2,5-dihydroxyterephthalaldehyde (DHTP-2CHO, 98%), were ordered from Sigma-Aldrich and Alfa Aesar.

### Synthesis of TPE-4NH<sub>2</sub>

BZP-2NH<sub>2</sub> (3.5 g, 15.73 mmol) was added to 170 mL of HCl at 60 °C. Subsequently, 10.5 g (88 mmol) of Sn was gradually added, and the reaction for 24 h at 80 °C. After cooling the flask, the white precipitate was rinsed with NaOH (1 M) solution to acquire a green powder (Scheme S1). FTIR (cm<sup>-1</sup>): 3423, 3359 (N–H), 3030 (aromatic C–H), 1616. <sup>1</sup>H NMR [DMSO-*d*<sub>6</sub>,  $\delta$ , ppm, Figure S1]: 6.6 (8H), 6.3 (8H), 4.8 (NH<sub>2</sub>). <sup>13</sup>C NMR [DMSO-*d*<sub>6</sub>,  $\delta$ , ppm, Figure S2]: 146–113.1.

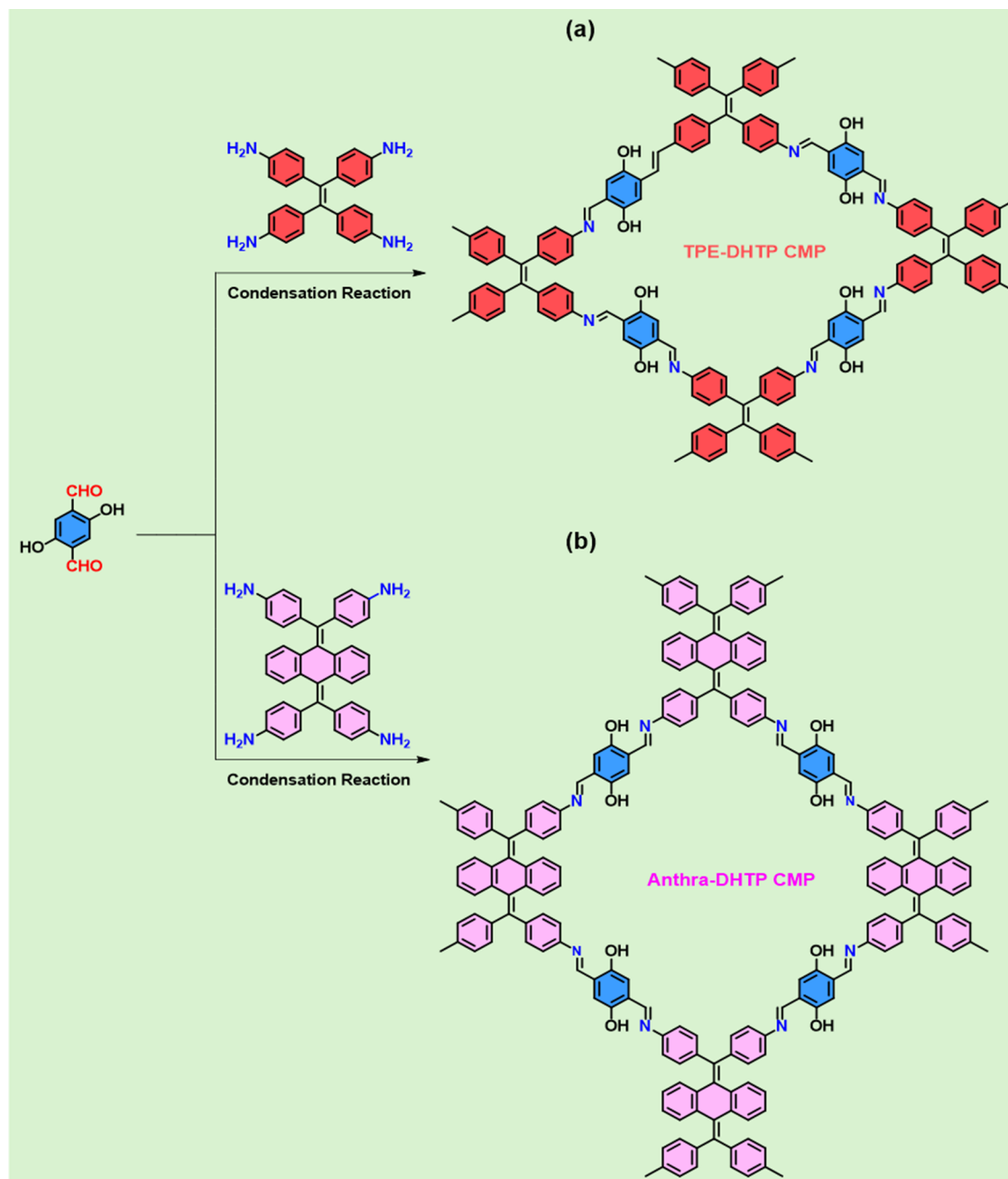
### Synthesis of

### 9,10-Bis(dibromomethylene)-9,10-dihydroanthracene (Anthra-Br<sub>4</sub>)

CBr<sub>4</sub> (80 g, 303.7 mmol), Anthra (10 g, 48 mmol), and PPh<sub>3</sub> (50 g, 151 mmol) in a 250 mL two-neck flask. Subsequently, we added 700 mL of toluene to the flask and cooled the mixture to –10 °C using an ice bath and acetone. The reaction mixture was kept for 2 h at –10 °C and then heated to 100 °C. The insoluble material was removed by vacuum filtration following the reaction. The organic layer was evaporated using a rotary evaporator, and the resulting substance was recrystallized from MeOH, yielding a light-yellow solid [6.37 g, Scheme S2]. FTIR (cm<sup>-1</sup>, Figure S3): 3069 (aromatic C–H), 1559. <sup>1</sup>H NMR [DMSO-*d*<sub>6</sub>,  $\delta$ , ppm, Figure S4]: 7.8–7.83 (4H), 7.3–7.3 (4H). <sup>13</sup>C NMR [DMSO-*d*<sub>6</sub>,  $\delta$ , ppm, Figure S5]: 139.8–91.45.

### Synthesis of Anthra-4Ph-4NH<sub>2</sub>

K<sub>2</sub>CO<sub>3</sub> (3.2 g, 23.1 mmol), Pd(PPh<sub>3</sub>)<sub>4</sub> (0.33 g, 0.3 mmol), Anthra-Br<sub>4</sub> (1.00 g, 2.9 mmol), BZB-NH<sub>2</sub> (3.2 g, 23.1 mmol), and were mixed in a 250 mL two-neck flask containing a combination of DO and H<sub>2</sub>O (80/40 mL) and increased the temperature reaction to 100 °C for 48 h. The insoluble material was removed by filtration and the reaction



**Figure 1.** Using a Schiff base reaction for the synthesis of (a) TPE-DHTP and (b) Anthra-DHTP CMPs from DHTP-2CHO as the building block.

solution was poured into HCl (5 mL) with H<sub>2</sub>O (400 mL) resulting in the formation of a light blue precipitate. This solid was then added to 300 mL of MeOH and stirred vigorously for 1 h at 60 °C, yielding Anthra-4Ph-4NH<sub>2</sub> (2.5 g, Scheme S3). FTIR (cm<sup>-1</sup>): 3429, 3344 (N–H), 3024 (aromatic C–H). <sup>1</sup>H NMR [DMSO-*d*<sub>6</sub>, δ, ppm, Figure S6]: 7.62, 6.95, 6.69, 6.44, 4.99 (NH<sub>2</sub>). 7.95–6.44. <sup>13</sup>C NMR [DMSO-*d*<sub>6</sub>, δ, ppm, Figure S7]: 147.9–113.82.

#### Preparation of TPE-DHTP CMP and Anthra-DHTP CMP

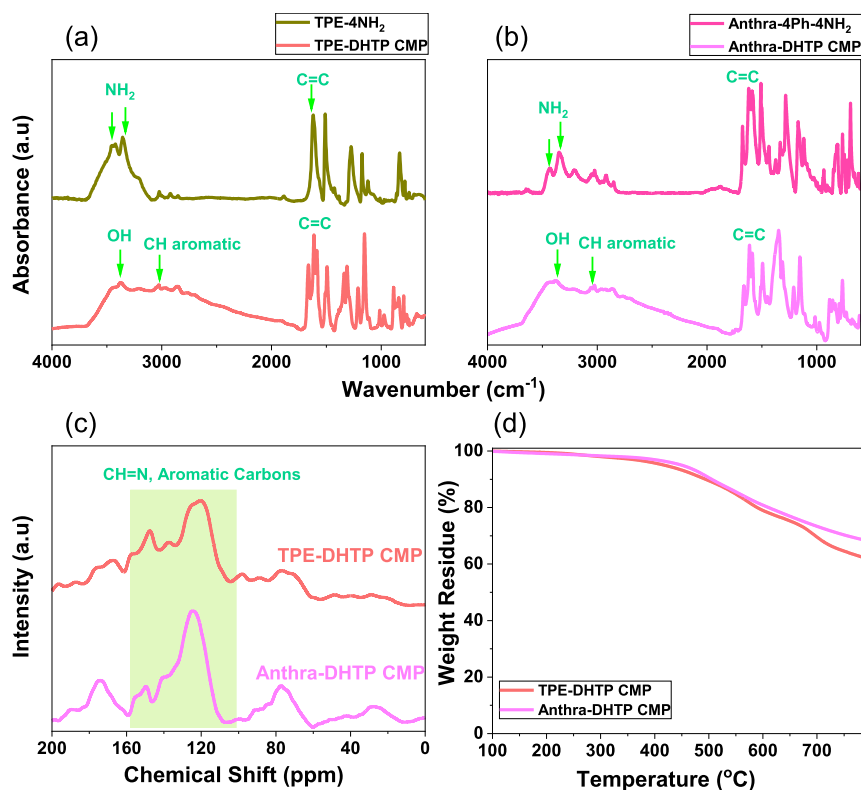
Use a 25 mL Schlenk tube, 0.10 g of TPE-4NH<sub>2</sub> (0.25 mmol), 0.08 g of DHTP-2CHO (0.48 mmol), DO/mesitylene (6 mL/6 mL), and 1.5 mL of AcOH (3 M). The solution mixture was refluxed at 110 °C for 3 days. After the reaction, the solid was filtered and purified using Soxhlet extraction [with THF and EtOH; respectively]. Finally, the resulting mixture was dried in a vacuum oven for 1 day, yielding a red powder known as TPE-DHTP CMP. FTIR (KBr, cm<sup>-1</sup>): 3373 (OH stretching), 3029 (C=C–H), 1663 (C=N). Solid-state <sup>13</sup>C NMR: 147.31 (C=N), 137.89–120.75 ppm (aromatic carbons). For

**Anthra-DHTP CMP:** 0.10 g of Anthra-4Ph-4NH<sub>2</sub> (0.18 mmol), 0.08 g of DHTP-2CHO (0.48 mmol), DO/mesitylene (6 mL/6 mL), and 1.5 mL of AcOH (3 M) to obtain Anthra-DHTP CMP as a brown powder (Yield: 84%) FTIR (cm<sup>-1</sup>): 3372 (OH stretching), 3037 (C=C–H), 1666 (C=N). Solid-state <sup>13</sup>C NMR: 149 (C=N), 140.12–124.57 ppm (aromatic carbons). As illustrated in Figure S8, the solubility of TPE-DHTP and Anthra-DHTP CMPs was tested in various organic solvents, including acetone, MeOH, DMF, DCM, and THF. The results indicated that both TPE-DHTP and Anthra-DHTP CMPs were insoluble in these solvents.

## RESULTS AND DISCUSSION

### Synthesis and Characterization of TPE-DHTP CMP and Anthra-DHTP CMP

We developed TPE-DHTP and Anthra-DHTP-linked CMPs (DHTP-CMPs) enriched with OH and C=N units, leveraging

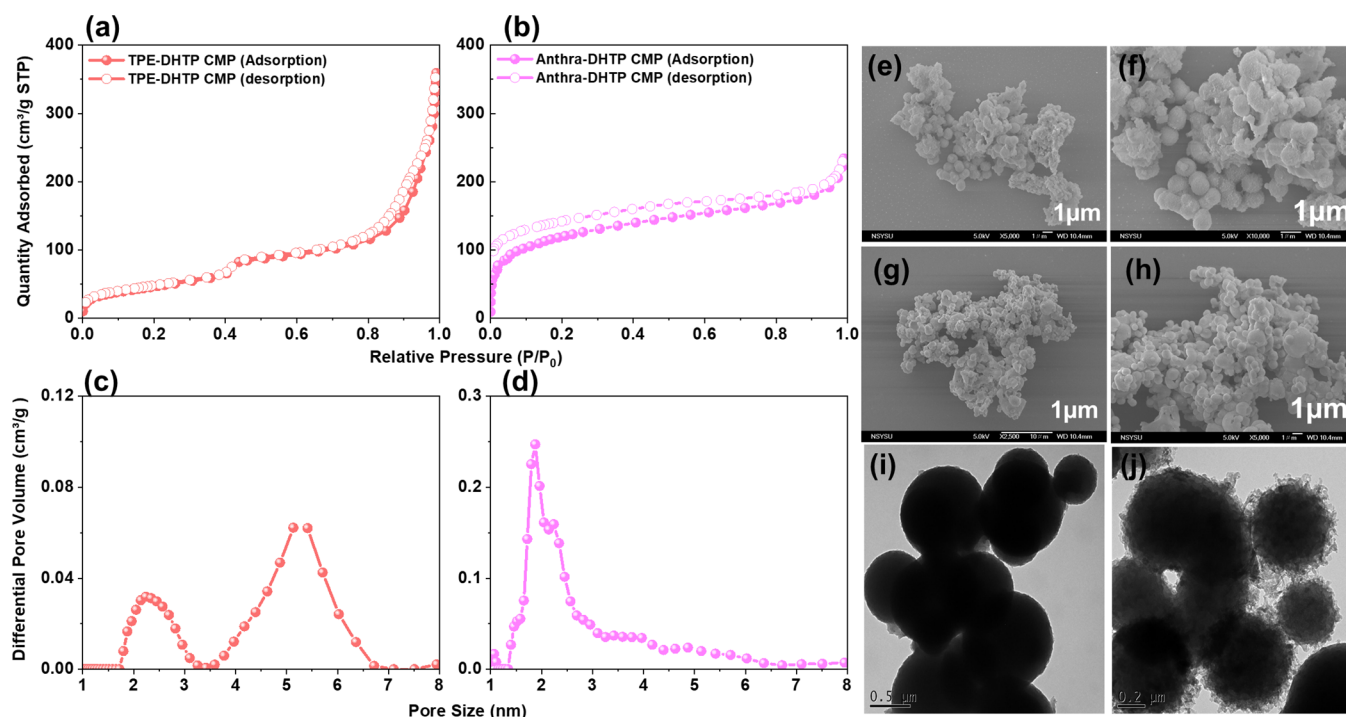


**Figure 2.** (a) and (b) FTIR spectra of TPE-4NH<sub>2</sub>, TPE-DHTP CMP, Anthra-4Ph-4NH<sub>2</sub>, and Anthra-DHTP CMP. (c) Solid-state <sup>13</sup>C NMR spectra and (d) TGA traces of TPE-DHTP and Anthra-DHTP CMPs.

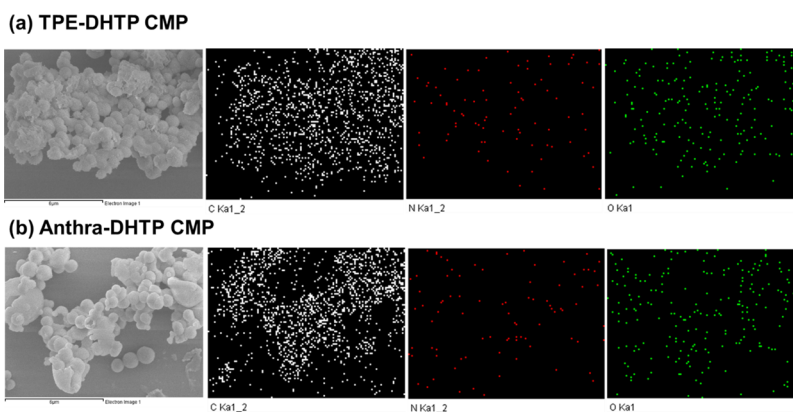
the potential of heteroatoms (N and O) in CMP structures for effective adsorption sites in supercapacitors and CO<sub>2</sub> capture [through robust physical interactions]. These DHTP-CMPs were synthesized using the Schiff base condensation polymerization technique [4 + 2]. Figure 1a,b illustrate the utilization of DHTP-2CHO [served as the building unit and a source of OH groups in the synthesis process] and TPE-4NH<sub>2</sub> and Anthra-4Ph-4NH<sub>2</sub> in DO/mesitylene with AcOH (3 M). Scheme S1 outlines the synthesis of TPE-4NH<sub>2</sub>, which is achieved by reacting BZP-2NH<sub>2</sub> with Sn in an HCl solution, yielding a green powder. In Scheme S2, Anthra-4Ph-4NH<sub>2</sub> is prepared by reacting the Anthra unit with CBr<sub>4</sub>/PPh<sub>3</sub> in toluene to obtain Anthra-Br<sub>4</sub>, a light-yellow powder. Subsequently, Scheme S3 illustrates the Suzuki coupling reaction between Anthra-Br<sub>4</sub> and BZB-NH<sub>2</sub> in the presence of Pd(PPh<sub>3</sub>)<sub>4</sub>/K<sub>2</sub>CO<sub>3</sub> in DO/H<sub>2</sub>O mixtures to produce Anthra-4Ph-4NH<sub>2</sub>. To confirm the synthesis of TPE-4NH<sub>2</sub>, Anthra-4Ph-4NH<sub>2</sub>, TPE-DHTP, and Anthra-DHTP-linked CMPs, FT-IR spectroscopy was initially conducted (Figure 2a,b). Strong absorption peaks corresponding to NH<sub>2</sub> groups, aromatic CH bonds, and C=C bonds are evident in Figure 2a,b. Specifically, for TPE-NH<sub>2</sub>, peaks were observed at 3423, 3359, 3030, and 1616 cm<sup>-1</sup>, while for Anthra-4Ph-4NH<sub>2</sub>, peaks appeared at 3429, 3344, 3024, and 1612 cm<sup>-1</sup>. The OH and aromatic CH absorption bands in the TPE-DHTP and Anthra-DHTP CMPs were detected between 3378 and 3372 cm<sup>-1</sup> and 3029 to 3037 cm<sup>-1</sup>, respectively, following the [4 + 2] condensation process involving DHTP-2CHO with TPE-4NH<sub>2</sub> and Anthra-4Ph-4NH<sub>2</sub>. Furthermore, the absence of the C=O and NH signals in FTIR spectra of DHTP-CMPs indicates the successful construction of the DHTP-CMPs.<sup>61,62</sup> The structure of both DHTP-CMPs was further analyzed using solid-state <sup>13</sup>C NMR spectroscopy. In the TPE-DHTP CMP,

two distinct resonance peaks were observed: one at 147.31 ppm corresponding to the C=N signal, and another in the range of 137.89–120.75 ppm representing aromatic carbon signals, as depicted in Figure 2c.

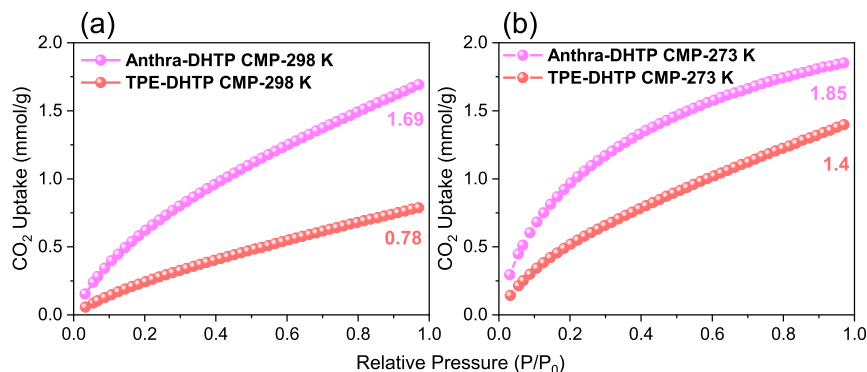
Similarly, in the Anthra-DHTP CMP, the C=N signal appeared at 149.0 ppm, while the aromatic carbon signals were observed in the range of 140.12–124.57 ppm. The XPS survey of TPE-DHTP and Anthra-DHTP CMPs, as seen in Figure S9a and S10a, revealed three peaks corresponding to the binding energies of C 1s, N 1s, and O 1s. The chemical states of C, N, and O for TPE-DHTP and Anthra-DHTP CMPs were further analyzed by deconvoluting the high-resolution XPS spectra. The three peaks of high-resolution C 1s XPS spectra deconvolution for TPE-DHTP and Anthra-DHTP CMPs are associated with C=C–C (284.4 eV), C–OH (286.3 eV), and C=N (285.38 eV), as shown in Figures S9b and S10b. Both TPE-DHTP and Anthra-DHTP CMPs displayed one peak for C=N–C bonds at 399.0 and 398.8 eV, respectively, based on the HR N 1s XPS spectral deconvolution, as presented in Figures S9c and S10c. Additionally, the O 1s XPS spectral deconvolution revealed a C–OH peak centered at 532.5 eV for TPE-DHTP CMP and at 532.8 eV for Anthra-DHTP CMP, as illustrated in Figures S9d and S10d. According to the TGA results (Figure 2d), TPE-DHTP CMP exhibited *T*<sub>d5</sub> and *T*<sub>d10</sub> values of 417 and 493 °C, respectively. Upon heating to 800 °C, it yielded a char yield of 62 wt %. Similarly, Anthra-DHTP CMP showed *T*<sub>d5</sub> and *T*<sub>d10</sub> values of 449 and 505 °C, respectively, with a char yield of 68 wt %. These findings underscore the excellent thermal stability of both DHTP-CMPs. Furthermore, Anthra-DHTP CMP exhibits superior thermal stability compared to TPE-DHTP CMP, likely due to the structural distinctions between the two compounds. The structure of Anthra-DHTP CMP may include stronger



**Figure 3.** (a and b)  $N_2$  adsorption–desorption isotherms and (c and d) pore size distributions of (a, c) TPE-DHTP CMP, and (b, d) Anthra-DHTP CMP. (e–h) SEM images and (i, j) TEM images of (e, f, i) TPE-DHTP CMP and (f, h, j) Anthra-DHTP CMP.



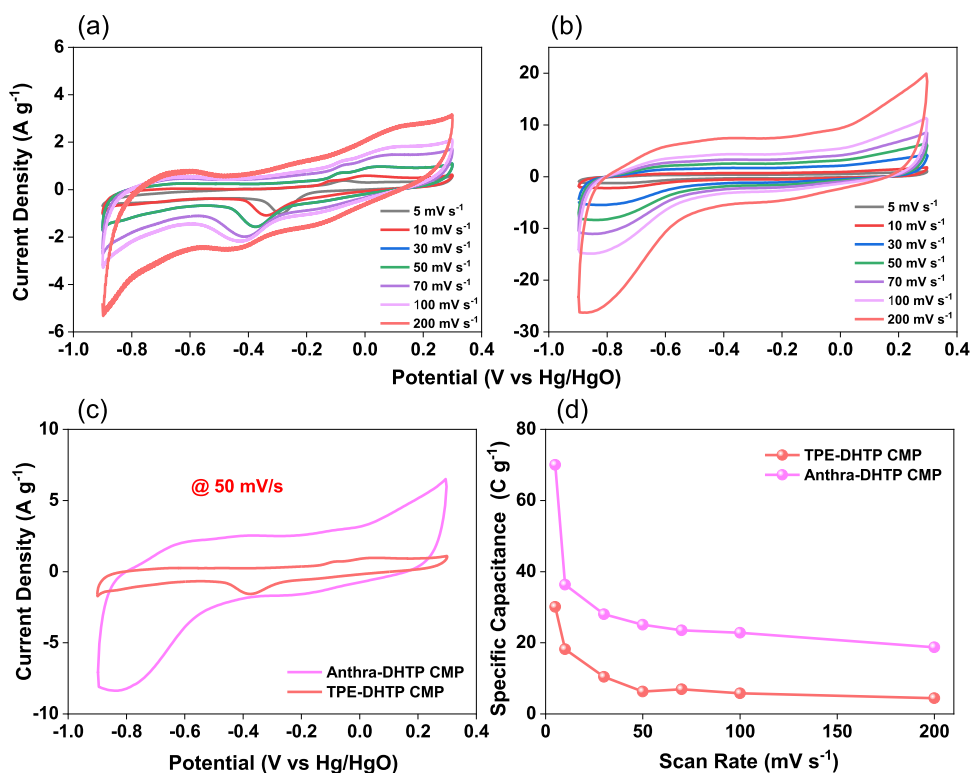
**Figure 4.** Elemental analyses through SEM-EDS mapping of (a) TPE-DHTP CMP and (b) Anthra-DHTP CMP.



**Figure 5.**  $CO_2$  absorption efficiency of the TPE-DHTP and Anthra-DHTP CMPs was evaluated at temperatures of (a) 298 and (b) 273 K.

intermolecular forces and a more rigid framework, contributing to its higher thermal resistance. We conducted  $N_2$  adsorption/desorption isotherm measurements at 77 K and utilized SEM and TEM to investigate the morphology (Figure 3), aiming to

elucidate the porous characteristics of DHTP-CMPs. Figure 3a,b depict the  $N_2$  adsorption/desorption isotherm curves. The results indicate that TPE-DHTP CMP exhibits a rapid increase in adsorption at low relative pressures ( $P/P_0 < 0.1$ ), indicative



**Figure 6.** Cyclic voltammetry (CV) curves of (a) TPE-DHTP and (b) Anthra-DHTP CMPs at 5–200  $\text{mV s}^{-1}$ , (c) CV curve of DHTP-CMPs electrodes were obtained using a scan rate of 50  $\text{mV s}^{-1}$  and (d) the relationship between specific capacitance and scan rate for DHTP-CMPs electrodes.

of strong  $\text{N}_2$  interaction. According to the IUPAC classification, this isotherm curve is categorized as Type II. Additionally, Brunauer–Emmett–Teller (BET) analysis revealed a total pore volume ( $V_{\text{total}}$ ) of  $0.53 \text{ cm}^3 \text{ g}^{-1}$  and a  $\text{BET}_{\text{SA}}$  of  $168 \text{ m}^2 \text{ g}^{-1}$ . The pore size distribution reveals two pore sizes at 2.3 and 4.2 nm (Figure 3c). Similarly, Anthra-DHTP CMP exhibits a Type IV isotherm characterized by a rapid increase in adsorption at low relative pressures and the presence of a hysteresis loop. This suggests its microporous nature, supported by  $\text{BET}_{\text{SA}}$  of  $431 \text{ m}^2 \text{ g}^{-1}$ , a  $V_{\text{total}}$  of  $0.36 \text{ cm}^3 \text{ g}^{-1}$ , and a pore size of 1.9 and 2.3 nm (Figure 3d). The TPE-DHTP-CMP and Anthra-DHTP-CMP exhibit irregular non-spherical particles, as shown by the SEM pictures in Figure 3e–h. The SEM findings were also comparable with TEM images (Figure 3i,j).

SEM-EDS element mapping images depicted in Figure 4a,b indicate the uniform distribution of elements C (white color), N (red color), and O (green color) throughout the networks of both TPE-DHTP CMP and Anthra-DHTP CMP. Figures S11 and S12 from the XRD study illustrate the amorphous nature of both DHTP-CMP networks, as evidenced by a broad peak.

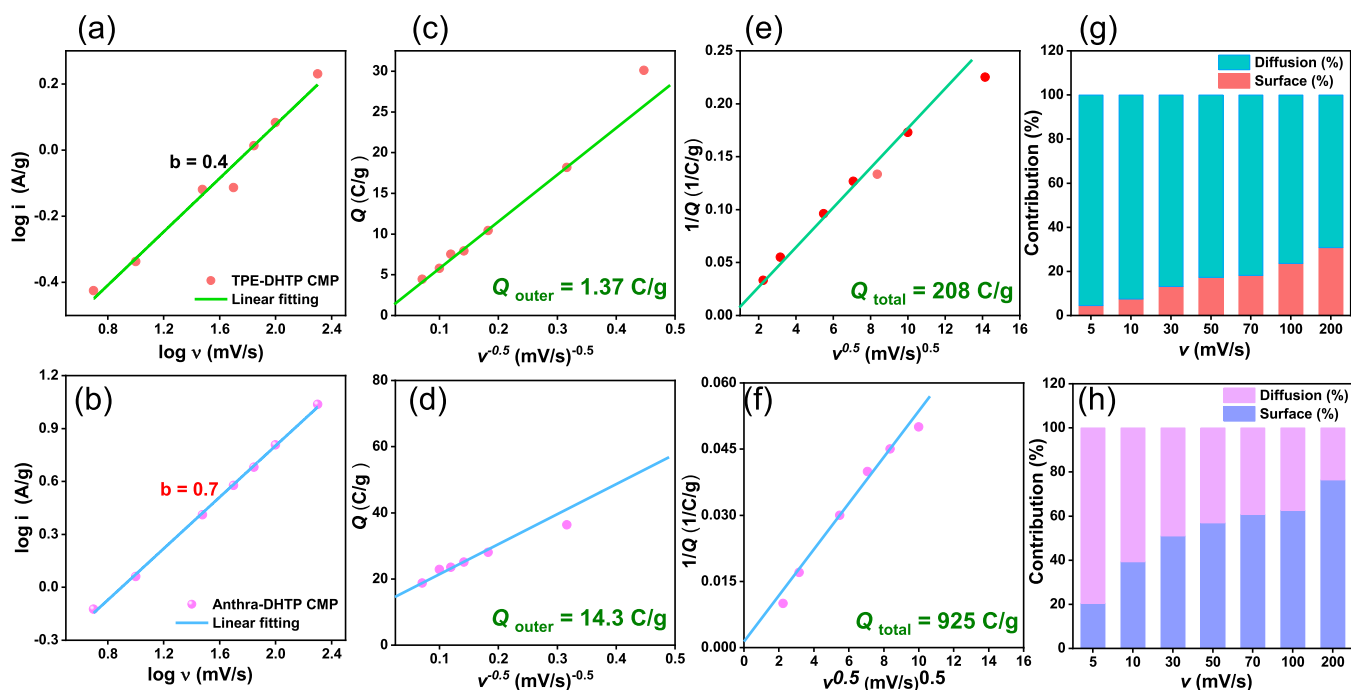
As seen in Figure 5a,b, further research was conducted on the  $\text{CO}_2$  uptake isotherms at 298 and 273 K for TPE-DHTP CMP and Anthra-DHTP CMP to evaluate their practical adoption. The Anthra-DHTP CMP exhibits a higher capacity for  $\text{CO}_2$  adsorption at both 298 K ( $1.69 \text{ mmol g}^{-1}$ , 7.43 wt %) and 273 K ( $1.85 \text{ mmol g}^{-1}$ , 8.14 wt %) compared to TPE-DHTP CMP [ $0.87$  (3.83 wt %) and  $1.4$  (6.16 wt %)  $\text{mmol g}^{-1}$ , respectively measured at 298 and 273 K]. The variation in adsorption can be ascribed to the greater  $\text{BET}_{\text{SA}}$  of Anthra-DHTP CMP and the potential presence of phenolic OH units and N atoms within its framework, which may act as basic sites

to capture acidic  $\text{CO}_2$  molecules.<sup>68–70</sup> The Clausius–Clapeyron equation was used to determine the isosteric heat of adsorption ( $Q_{\text{st}}$ ) for Anthra-DHTP and TPE-DHTP CMPs. The results revealed that at a low  $\text{CO}_2$  adsorption level of approximately  $0.5 \text{ mmol/g}$ , the  $Q_{\text{st}}$  for Anthra-DHTP CMP is  $29.01 \text{ kJ/mol}$ , while for TPE-DHTP CMP, it is  $21.46 \text{ kJ/mol}$ . These values were derived from the  $\text{CO}_2$  adsorption results obtained at 298 and 273 K. It is worth noting that our DHTP-CMPs demonstrate superior  $\text{CO}_2$  adsorption compared to other porous organic materials. These materials include pyrene-PAFs ( $0.90$ – $1.15 \text{ mmol g}^{-1}$ ),<sup>71</sup> CMP-1-(OH)<sub>2</sub> ( $1.8 \text{ mmol g}^{-1}$ ),<sup>72</sup> tri(4-ethynylphenyl)amine-PAFs ( $1.19 \text{ mmol g}^{-1}$ ),<sup>73</sup> and CMP-1-AMD1 ( $1.51 \text{ mmol g}^{-1}$ ).<sup>74</sup>

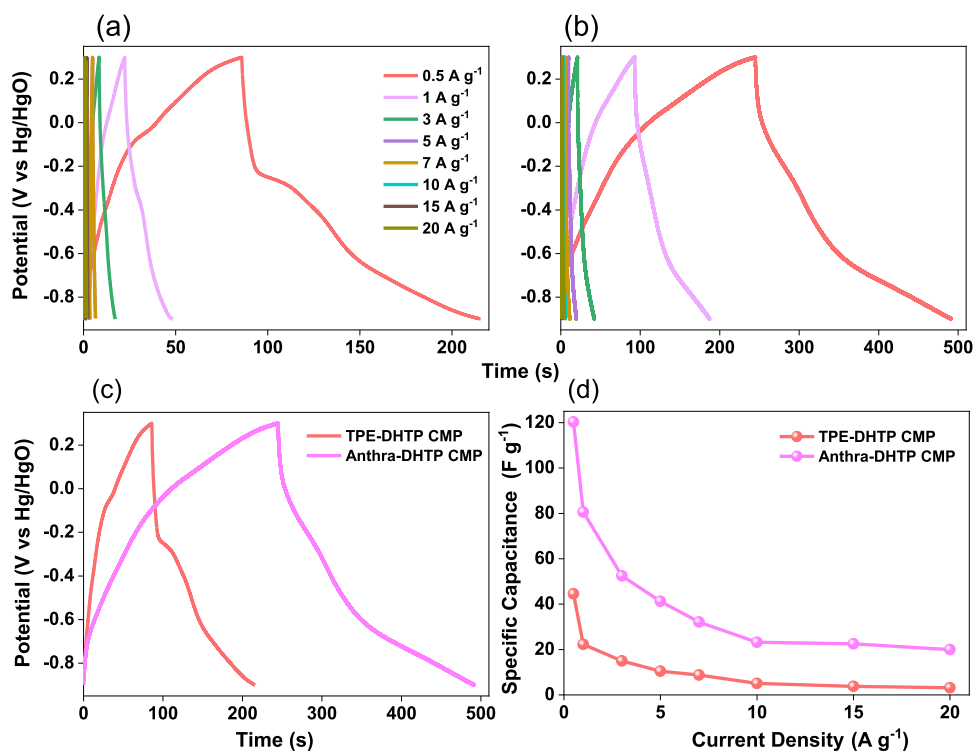
#### Electrochemical Performance of TPE-DHTP CMP and Anthra-DHTP CMP Based on Three Electrodes System

Using KOH (1 M) as the electrolyte in a three-electrode system, we initially investigated the electrochemical performance and mechanism of TPE-DHTP CMP and Anthra-DHTP CMP electrodes for supercapacitor applications. Our electrodes, made from these materials, were shaped like rectangles. The CV curves displayed humps over a potential window that varied with scan rate [Figure 6a,b]. As the potential scan rate increased, the peak current of both TPE-DHTP CMP and Anthra-DHTP CMP electrodes also rose. This indicated a small amount of charge transfer resistance, pseudocapacitance, and electric double-layer capacitance (EDLC) as the sources of their capacitive responses.<sup>20,75</sup> The electron-rich phenyl rings and heteroatoms contributed to this characteristic, as evidenced by the humps

in the rectangular CV curves.<sup>76</sup> The capacitive properties of Anthra-DHTP CMP are outstanding even at high scan rates, as presented by the CV curves for each electrode created at 50



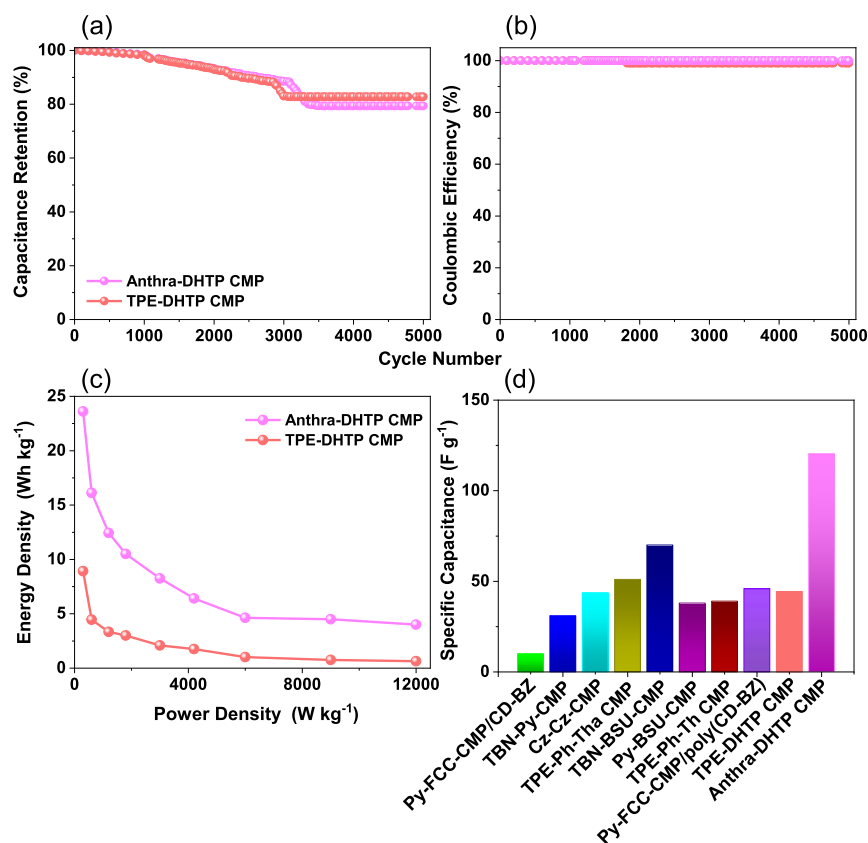
**Figure 7.** (a, b) Power's law relation, (c, d) relation between  $Q$  ( $\text{C g}^{-1}$ ) and  $v^{-0.5}$  ( $\text{mV s}^{-1}$ ) $^{-0.5}$ , (e, f)  $1/Q$  vs  $v^{0.5}$  ( $\text{mV s}^{-1}$ ) $^{0.5}$  for (a, c, e) TPE-DHTP and (b, d, f) Anthra-DHTP CMPs electrodes. Percentage of surface contribution and diffusion contribution for (g) TPE-DHTP and (h) Anthra-DHTP CMPs electrodes.



**Figure 8.** GCD curves of (a) TPE-DHTP and (b) Anthra-DHTP CMPs, (c) CGD curve of DHTP-CMPs electrodes were obtained using  $0.5 \text{ A g}^{-1}$  and (d) specific capacitances for DHTP-CMPs electrodes were calculated from GCD.

$\text{mV s}^{-1}$ , shown in Figure 6c. The CV curve of Anthra-DHTP CMP, which has a larger integral area compared to that of TPE-DHTP CMP, indicates the best capacity for charge storage. Consequently, as illustrated in Figure 6d, Tables S1 and S2, the distinctive capacitance of Anthra-DHTP CMP is greater than that of TPE-DHTP CMP at various scan rates.

The specific capacitance of Anthra-DHTP CMP is calculated to be 70, 36.34, 28, 25.1, 23.51, 22.83, and  $18.8 \text{ C g}^{-1}$  at 5, 10, 30, 50, 70, 100, and 200  $\text{mV s}^{-1}$ , respectively. Cyclic voltammetry (CV) is considered the most significant methodology for examining a material's response to a voltage range, as it provides insight into its electrochemical behavior, voltage



**Figure 9.** (a) The stability test, (b) Coulombic efficiency, (c) Ragone plot for Anthra-DHTP and TPE-DHTP CMPs, and (d) specific capacitances of the Anthra-DHTP and TPE-DHTP CMPs and those of other CMP materials previously reported for supercapacitors.

procedure, reversible nature (diffusion vs surface control), and energy storage mechanism. After reviewing the electrochemical findings, the capacitive aspect of total charge storage was investigated. The power law was utilized to analyze the electrodes' charge storage capability. This analysis can be expressed either as  $\log i = a + b \log v$  or  $i(V) = av^b$ , where  $i$  represents the applied current density and  $v$  the applied potential window. The intercept and slope of the  $\log i$  vs  $\log v$  plots are used to determine the parameters  $a$  and  $b$ , which are constants. The value of  $b$  is obtained from the slope of the linear fit of the  $\log i$  vs  $\log v$  plot at a fixed voltage. Figure 7a,b depict the plot of  $\log i$  versus  $\log v$ , showing that the  $b$  value ( $\sim 0.4$ ) is closer to 0.5. This indicates that ion intercalation is the dominant mechanism for charge storage. Conversely, a  $b$  value higher than 0.5 suggests that under these conditions, the capacitive contribution is more prevalent than the intercalation process. Additionally, the kinetics of charge storage in TPE-DHTP CMP and Anthra-DHTP CMP have been further examined using the Trasatti technique. Two different charge storage processes are found in this analysis: the buildup of charge on the outside surface of particles (capacitive) is frequently referred to as  $Q_{\text{outer}}$  (outer surface) and the first is named  $Q_{\text{bulk}}$  (inner surface).

$$Q = Q_{\text{outer}} + Q_{\text{bulk}}$$

Figure 7c,d show the relation among capacity  $Q$  and  $(v)^{-0.5}$ , where  $(v)$  is the possible scan rate. By finding an intercept of the graph of  $Q$  and  $(v)^{-0.5}$ , it can be done to determine the value of  $(Q_{\text{outer}})$  by the equation:

$$Q = Q_{\text{outer}} + Kv^{-0.5}$$

where  $K$  is a constant,  $v$  ( $\text{mV s}^{-1}$ ) is the potential scan rate, and  $Q$  ( $\text{C g}^{-1}$ ) is the capacity produced by every single CV sequence. It is determined that the maximum sweep-rate capacitance of TPE-DHTP CMP and Anthra-DHTP CMP is 1.37 and 14.3  $\text{C g}^{-1}$ .

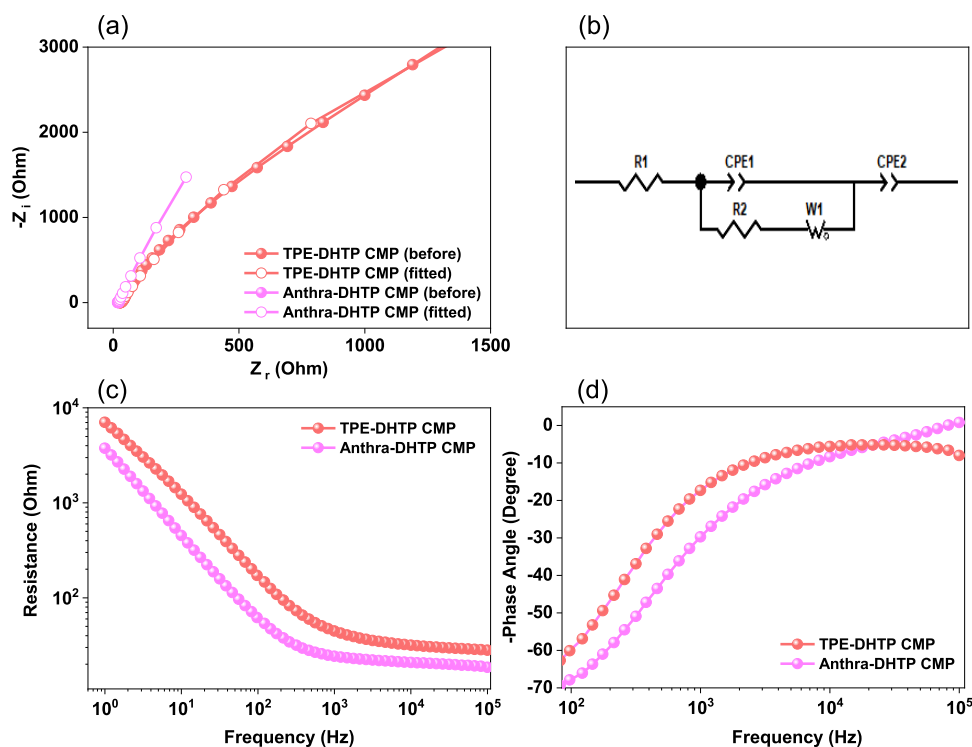
This figure represents the stored charge because of double-layer and/or pseudocapacitive capacitive mechanisms. This process takes place when the potential scan rate gets to the lowest possible levels, giving the ions sufficient time to disperse. Plotting  $1/Q$  against  $(v)^{0.5}$  yields the total charge ( $Q_{\text{total}}$ ), as seen in Figure 7e,f. Based on the relationship:

$$\frac{1}{Q} = \frac{1}{Q_{\text{total}}} + Kv^{0.5}$$

However, in this case, we can calculate the stored charge which is 208 and 952  $\text{C g}^{-1}$  for TPE-DHTP CMP and Anthra-DHTP CMP, respectively. The charge stored increases based on previously published techniques when the surface contribution rises, and diffusion limitation takes precedence over the scan rate. The percentage of surface and diffusion-controlled contributions at scan rates are thus displayed in Figure 7g,h.

Diffusion-controlled faradic yields are 95.4 and 79.5% for TPE-DHTP CMP and Anthra-DHTP CMP electrodes. In Figure 8a,b, TPE-DHTP CMP and Anthra-DHTP CMP electrodes underwent galvanostatic charge–discharge (GCD) tests. The resulting GCD curves, which were modestly bent and triangular, suggested characteristics of both pseudocapacitance and EDLC, consistent with the CV curve data. The noticeable potential drop at the onset of each discharge curve in the GCD profiles for the TPE-DHTP CMP and Anthra-





**Figure 10.** EIS curves: (a) Nyquist plots, (b) equivalent fitted circuit, (c) Bode plot of frequency-dependent resistance (magnitude), and (d) Bode plot of frequency-dependent phase angles of TPE-DHTP and Anthra-DHTP CMPs.

DHTP CMP electrodes (Figure 8a,b) can be attributed to the internal resistance within the electrode material.<sup>76,77</sup> Figure 8c compares the GCD curves of DHTP-CMPs at  $0.5 \text{ A g}^{-1}$ . It is observed that the Anthra-DHTP CMP materials exhibited longer discharge durations than the TPE-DHTP CMP materials. This suggests excellent capacitive behavior and good reversibility of charge/discharge cycles, indicating that the capacitance of the Anthra-DHTP CMP material was significantly enhanced. We determined the capacitances of the TPE-DHTP and Anthra-DHTP CMPs samples from their GCD profiles.  $0.5 \text{ A g}^{-1}$ , the specific capacitance of Anthra-DHTP CMP is  $121 \text{ F g}^{-1}$ , while that of TPE-DHTP CMP is  $44 \text{ F g}^{-1}$  (Figure 8c). According to the GCD data (Figure 8d), the specific capacitances of Anthra-DHTP CMP were 121, 81, 62, 53, 41, 32, 23, 23, and  $20 \text{ F g}^{-1}$ , respectively [measured at 0.5, 1, 2, 3, 5, 7, 10, 15, and  $20 \text{ A g}^{-1}$ ] are which aligns with the outcomes of the CV tests.

Continuous cycling experiments are valuable for assessing the longevity of electrode materials. After undergoing 5000 consecutive charge–discharge cycles, the capacitance stability ( $C_s$ ) of Anthra-DHTP CMP and TPE-DHTP CMP samples maintained 79 and 82% of their original capacitance and their Coulombic efficiency remained close to 100%, as shown in Figure 9a,b. The existence of heteroatoms and increased surface area are responsible for these results, which enhance the electrode–electrolyte interaction, and facilitate the flow of electrolyte ions.

Furthermore, the energy densities of the materials might be determined using the Ragone plot (Figure 9c). At specific power (SP) of  $300 \text{ W kg}^{-1}$ , for both DHTP-CMPs materials, indicating energy densities of 8.9 and  $23.6 \text{ Wh kg}^{-1}$  for TPE-DHTP and Anthra-DHTP CMPs. Anthra-DHTP CMP exhibits a maximum energy density of  $23.6 \text{ Wh kg}^{-1}$ , attributed to its enhanced surface area and porosity. These results are

promising and show the material's potential for SC applications. Table S3 presents an examination of DHTP CMPs electrodes, as well as other porous materials that have been reported to be employed in supercapacitor applications (Figure 9d). In this study, we utilized KOH electrolyte (1 M) to conduct Electrochemical Impedance Spectroscopy (EIS) measurements on our DHTP-CMPs. Table S4 presents the fitted data, which allowed us to generate Nyquist plots and corresponding electrical equivalent circuits for Anthra-DHTP CMP and TPE-DHTP CMP, depicted in Figure 10a,b. The EIS data was fitted to circuit models that included parameters such as  $Z_w$  (Warburg element),  $R_{ct}$  (charge transfer resistance),  $R_s$  (series resistance), and two constant phase elements (CPE-EDL, CPE-P). The initial  $R_s$  values for TPE-DHTP CMP and Anthra-DHTP electrodes were recorded at 30.7 and  $20.9 \text{ } \Omega$ , respectively [Table S4]. These relatively low ohmic resistance values indicate their superior capacitance. Additionally, Figure 10c illustrates the frequency-dependent magnitude Bode plot, highlighting the exceptional capacitive behavior of these DHTP-CMPs and emphasizing their potential in energy applications. Furthermore, Figure 10d displays the frequency-dependent phase angle Bode plot, revealing knee frequencies that act as metrics for evaluating the performance rate of electrode materials. The moderate knee frequencies observed for TPE-DHTP CMP and Anthra-DHTP CMP suggest that these DHTP-CMPs could be effective electrodes for various energy-related applications, demonstrating both capacitive and resistive characteristics. Figure S13 displays the results of EIS analysis for the DHTP-CMPs after 5000 cycles, presented through Nyquist plots across a frequency range from 100 kHz to 10 mHz. As observed, there was an observed increase in ion diffusion resistance, indicating a slowdown in ion diffusion and the slope of the straight line in the low-frequency region decreased

(showing a less vertical line), suggesting an increase in ion diffusion resistance. Additionally, after 5000 cycles, the semicircles in the high-frequency region (indicating charge transfer resistance) became more pronounced. These observations imply a potential loss of contact between the substrate and active material over multiple cycles. Based on these findings, Anthra-DHTP CMP demonstrated superior cycling performance and reversibility compared to TPE-DHTP CMP.

## CONCLUSIONS

In conclusion, we successfully synthesized two distinct forms of DHTP-CMPs featuring CH=N and phenolic units, TPE-DHTP CMP and Anthra-DHTP CMP, using the efficient and simple Schiff-base condensation reaction. Both DHTP-CMPs exhibit good thermal stability and porous properties, with Anthra-DHTP CMP standing out for its exceptional char yield: 68 wt % and  $T_{d10} = 505$  °C. Compared to TPE-DHTP CMP, Anthra-DHTP CMP possesses the largest  $BET_{SA}$  of  $431 \text{ m}^2 \text{ g}^{-1}$  and the highest  $\text{CO}_2$  absorption capacity at 273 K of  $1.85 \text{ mmol g}^{-1}$  (8.14 wt %). In electrochemical testing, TPE-DHTP CMP and Anthra-DHTP CMP showed capacitance values of 44 and  $121 \text{ F g}^{-1}$ , respectively, at  $0.5 \text{ A g}^{-1}$ , attributed to the behavior of their heteroatoms and excellent porosity. Therefore, these results confirm that Anthra-DHTP CMP is an ideal material choice for energy storage and gas capture applications. This work demonstrates that the efficient condensation process used to produce DHTP-CMPs has applications extending beyond  $\text{CO}_2$  absorption and supercapacitors. Additionally, these materials show potential for use in photocatalysis, iodine and dye capture, and other related fields.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/jacsau.4c00537>.

Additional experimental details, characterizations, electrochemical analysis, measurements, and schematic diagrams of TPE-4NH<sub>2</sub>, Anthra-Br<sub>4</sub>, and Anthra-4Ph-4NH<sub>2</sub>, along with FTIR and NMR data for TPE-4NH<sub>2</sub>, Anthra-Br<sub>4</sub>, and Anthra-4Ph-4NH<sub>2</sub>; data on solubility, XPS, XRD, and EIS for TPE-DHTP CMP and Anthra-DHTP CMP; as well as a performance comparison of supercapacitors using TPE-DHTP CMP and Anthra-DHTP CMP electrodes with previously described electrodes (PDF)

## AUTHOR INFORMATION

### Corresponding Author

**Shiao-Wei Kuo** – Department of Materials and Optoelectronic Science, Center for Functional Polymers and Supramolecular Materials, National Sun Yat-Sen University, Kaohsiung 804, Taiwan; Department of Medicinal and Applied Chemistry, Kaohsiung Medical University, Kaohsiung 807, Taiwan; [orcid.org/0000-0002-4306-7171](https://orcid.org/0000-0002-4306-7171); Email: [kuosw@faculty.nsysu.edu.tw](mailto:kuosw@faculty.nsysu.edu.tw)

### Authors

**Mohamed Gamal Mohamed** – Department of Materials and Optoelectronic Science, Center for Functional Polymers and Supramolecular Materials, National Sun Yat-Sen University, Kaohsiung 804, Taiwan; Department of Chemistry, Faculty

of Science, Assiut University, Assiut 71516, Egypt;

[orcid.org/0000-0003-0301-8372](https://orcid.org/0000-0003-0301-8372)

**Chia-Chi Chen** – Department of Materials and Optoelectronic Science, Center for Functional Polymers and Supramolecular Materials, National Sun Yat-Sen University, Kaohsiung 804, Taiwan

**Mervat Ibrahim** – Chemistry Department, Faculty of Science, New Valley University, El-Kharja 72511, Egypt

**Aya Osama Mousa** – Department of Materials and Optoelectronic Science, Center for Functional Polymers and Supramolecular Materials, National Sun Yat-Sen University, Kaohsiung 804, Taiwan

**Mohamed Hammad Elsayed** – Department of Chemistry, Faculty of Science, Al-Azhar University, Cairo 11884, Egypt; [orcid.org/0000-0001-7462-0294](https://orcid.org/0000-0001-7462-0294)

**Yunsheng Ye** – Department of Materials and Optoelectronic Science, Center for Functional Polymers and Supramolecular Materials, National Sun Yat-Sen University, Kaohsiung 804, Taiwan

Complete contact information is available at: <https://pubs.acs.org/10.1021/jacsau.4c00537>

## Author Contributions

CRedit: **Mohamed Gamal Mohamed** conceptualization, data curation, formal analysis, investigation, methodology, writing-original draft, writing-review & editing; **Chia-Chi Chen** conceptualization, data curation, investigation; **Mervat Ibrahim** conceptualization, investigation; **Aya Osama Mousa** data curation; **Mohamed Hammad Elsayed** conceptualization; **Yunsheng Ye** conceptualization, formal analysis; **Shiao-Wei Kuo** funding acquisition, project administration, supervision.

## Notes

The authors declare no competing financial interest.

## ACKNOWLEDGMENTS

This study was supported financially by the National Science and Technology Council, Taiwan, under contracts NSTC 112-2223-E-110-002- and 112-2218-E-110-007. The authors thank the staff at National Sun Yat-sen University for their assistance with the TEM (ID: EM022600) experiments.

## REFERENCES

- (1) Chu, H.; Huang, Z.; Zhang, Z.; Yan, X.; Qiu, B.; Xu, N. Integration of carbon emission reduction policies and technologies: Research progress on carbon capture, utilization and storage technologies. *Sep. Purif. Technol.* **2024**, 343, No. 127153.
- (2) Mousa, A. O.; Mohamed, M. G.; Chuang, C. H.; Kuo, S. W. Carbonized Amino-Linked Porous Organic Polymers Containing Pyrene and Triazine Units for Gas Uptake and Energy Storage. *Polymers* **2023**, 15, 1891.
- (3) Saighani, A.; Sommer, C. Potentials for Reducing Carbon Dioxide Emissions and Conversion of Renewable Energy for the Regional Transport Market—A Case Study. *Transp. Res. Procedia* **2017**, 25, 3479–3494.
- (4) Mohamed, M. G.; Hu, H. Y.; Madhu, M.; Samy, M. M.; Mekhemer, I. M. A.; Tseng, W. L.; Chou, H. H.; Kuo, S. W. Ultrastable Two-Dimensional Fluorescent Conjugated Microporous Polymers Containing Pyrene and Fluorene Units for Metal Ion Sensing and Energy Storage. *Eur. Polym. J.* **2023**, 189, No. 111980.
- (5) Piantini, R. D.; Mujumdar, A. S. Climate Change and Drying of Agricultural Products. *Dry. Technol.* **2009**, 27, 629–635.

- (6) Li, Y.; Yang, X.; Du, E.; Liu, Y.; Zhang, S.; Yang, C.; Zhang, N.; Liu, C. A review on carbon emission accounting approaches for the electricity power industry. *Appl. Energy* **2024**, *359*, No. 122681.
- (7) Mohamed, M. G.; Chang, W. C.; Kuo, S. W. Crown Ether- and Benzoxazine-Linked Porous Organic Polymers Displaying Enhanced Metal Ion and CO<sub>2</sub> Capture through Solid State Chemical Transformation. *Macromolecules* **2022**, *55*, 7879–7892.
- (8) Ray, B.; Churipard, S. R.; Peter, S. C. An overview of the materials and methodologies for CO<sub>2</sub> capture under humid conditions. *J. Mater. Chem. A* **2021**, *9*, 26498–26527.
- (9) Lin, M. C.; Kuo, S. W.; Mohamed, M. G. High-Performance Anthracene-Linked Covalent Triazine Framework with Dual Functions for CO<sub>2</sub> Capture and Supercapacitor Applications. *Mater. Adv.* **2024**, *5*, 6222.
- (10) Liu, C.; Bai, Y.; Li, W.; Yang, F.; Zhang, G.; Pang, H. In situ growth of three-dimensional MXene/metal–organic framework composites for high-performance supercapacitors. *Angew. Chem., Int. Ed.* **2022**, *61*, No. e202116282.
- (11) Mohamed, M. G.; Elsayed, M. H.; Hassan, A. E.; Basit, A.; Mekhemer, I. M. A.; Chou, H. H.; Chen, K. H.; Kuo, S. W. Hybrid Porous Polymers Combination of Octavinylsilsesquioxane/Pyrene with Benzothiadiazole Units for Robust Energy Storage and Efficient Photocatalytic Hydrogen Production from Water. *ACS Appl. Polym. Mater.* **2024**, *6*, 5945–5956.
- (12) Osama Mousa, A.; Sharma, S. U.; Chaganti, S. V.; Mansoure, T. H.; Singh, P. N.; Ejaz, M.; Chuang, C. H.; Lee, J. T.; Kuo, S. W.; Mohamed, M. G. Designing strategically functionalized conjugated microporous polymers with pyrene and perylenetetra-carboxylic dianhydride moieties with single-walled carbon nanotubes to enhance supercapacitive energy storage efficiency. *J. Power Sources* **2024**, *608*, No. 234624.
- (13) Wang, S. H.; Li, L.; Wang, H.; Wang, X. W.; Wang, T. H. Continuous and Scalable Manufacture of Coal-Derived Hierarchical Porous Carbon Dominated with Mesopores for High Rate-Performance Supercapacitors. *ACS Appl. Energy Mater.* **2024**, *7*, 4268–4278.
- (14) Mohamed, M. G.; Su, B. X.; Kuo, S. W. Robust Nitrogen-Doped Microporous Carbon via Crown Ether-Functionalized Benzoxazine-Linked Porous Organic Polymers for Enhanced CO<sub>2</sub> Adsorption and Supercapacitor Applications. *ACS Appl. Mater. Interfaces* **2024**, *16*, 40858.
- (15) Yu, Z.; Tetard, L.; Zhai, L.; Thomas, J. Supercapacitor electrode materials: Nanostructures from 0 to 3 dimensions. *Energy Environ. Sci.* **2015**, *8*, 702–730.
- (16) Singh, P. N.; Mohamed, M. G.; Kuo, S. W. Systematic Design and Synthesis of Conjugated Microporous Polymers Containing Pyrene and Azobenzene Building Materials for High-Performance Energy Storage. *ACS Appl. Energy Mater.* **2023**, *6*, 11342–11351.
- (17) Libich, J.; Máca, J.; Vondrák, J.; Čech, O.; Sedlářková, M. Supercapacitors: Properties and applications. *J. Energy Storage* **2018**, *17*, 224–227.
- (18) Ejaz, M.; Mohamed, M. G.; Chen, Y. T.; Zhang, K.; Kuo, S. W. Porous carbon materials augmented with heteroatoms derived from hyperbranched biobased benzoxazine resins for enhanced CO<sub>2</sub> adsorption and exceptional supercapacitor performance. *J. Energy Storage* **2024**, *78*, No. 110166.
- (19) Mohamed, M. G.; Sharma, S. U.; Liu, N. Y.; Mansoure, T. H.; Samy, M. M.; Chaganti, S. V.; Chang, Y. L.; Lee, J. T.; Kuo, S. W. Ultrastable covalent triazine organic framework based on anthracene moiety as a platform for high-performance carbon dioxide adsorption and supercapacitors. *Int. J. Mol. Sci.* **2022**, *23*, 3174.
- (20) Mei, L.; Cui, X.; Duan, Q.; Li, Y.; Lv, X.; Wang, H. Metal phthalocyanine-linked conjugated microporous polymer hybridized with carbon nanotubes as a high-performance flexible electrode for supercapacitors. *Int. J. Hydrogen Energy* **2020**, *45*, 22950–22958.
- (21) Mohamed, M. G.; Chaganti, S. V.; Li, M. S.; Samy, M. M.; Sharma, S. U.; Lee, J. T.; Elsayed, M. H.; Chou, H. H.; Kuo, S. W. Ultrastable Porous Organic Polymers Containing Thianthrene and Pyrene Units as Organic Electrode Materials for Supercapacitors. *ACS Appl. Energy Mater.* **2022**, *5*, 6442–6452.
- (22) Mohamed, M. G.; Ahmed, M. M.; Du, W. T.; Kuo, S. W. Meso/Microporous Carbons on Conjugated Hyper-Crosslinked Polymers Based on Tetraphenylethene for High-Performance CO<sub>2</sub> Capture and Supercapacitor. *Molecules* **2021**, *26*, 738.
- (23) Zhao, M.; Zhao, Q.; Li, B.; Xue, H.; Pang, H.; Chen, C. Recent progress in layered double hydroxide-based materials for electrochemical capacitors: Design, synthesis and performance. *Nanoscale* **2017**, *9*, 15206–15225.
- (24) Weng, T. H.; Mohamed, M. G.; Sharma, S. U.; Mekhemer, I. M. A.; Chou, H. H.; Kuo, S. W. Rationally Engineered Ultrastable Three-Dimensional (3D) Conjugated Microporous Polymers Containing Triptycene, Tetraphenylethene, and Benzothiadiazole Units as Exceptional High-Performance Organic Electrodes for Supercapacitors. *ACS Appl. Energy Mater.* **2023**, *6*, 9012–902457.
- (25) Winter, M.; Brodd, R. J. What are batteries, fuel cells, and supercapacitors. *Chem. Rev.* **2004**, *104*, 4245–4269.
- (26) Muzaffar, A.; Ahamed, M. B.; Deshmukh, K.; Thirumalai, J. A review on recent advances in hybrid supercapacitors: Design, fabrication and applications. *Renew. Sustain. Energy Rev.* **2019**, *101*, 123–145.
- (27) Mohamed, M. G.; Samy, M. M.; Mansoure, T. H.; Li, C. J.; Li, W. C.; Chen, J. H.; Zhang, K.; Kuo, S. W. Microporous Carbon, and Carbon/Metal Composite Materials Derived from Bio-Benzoxazine-Linked Precursor for CO<sub>2</sub> Capture and Energy Storage Applications. *Int. J. Mol. Sci.* **2022**, *23*, 347.
- (28) Zhang, X.; Qu, G.; Wang, Z.; Xiang, G.; Hao, S.; Wang, X.; Xu, X.; Ma, W.; Zhao, G. Hollow polyhedron structure of amorphous Ni-Co-S/Co(OH)<sub>2</sub> for high performance supercapacitors. *Chin. Chem. Lett.* **2021**, *32*, 2453–2458.
- (29) Samy, M. M.; Gamal Mohamed, M.; Sharma, S. U.; Chaganti, S. V.; Hassan Mansoure, T.; Lee, J. T.; Chen, T.; Kuo, S. W. Constructing conjugated microporous polymers containing triphenylamine moieties for high-performance capacitive energy storage. *Polymer* **2023**, *264*, No. 125541.
- (30) Samy, M. M.; Mohamed, M. G.; Kuo, S. W. Directly synthesized nitrogen-and-oxygen-doped microporous carbons derived from a bio-derived polybenzoxazine exhibiting high-performance supercapacitance and CO<sub>2</sub> uptake. *Eur. Polym. J.* **2020**, *138*, No. 109954.
- (31) Zhang, L.; Zhao, X. S. Carbon-based materials as supercapacitor electrodes. *Chem. Soc. Rev.* **2009**, *38*, 2520–2531.
- (32) Mohamed, M. G.; Chang, S. Y.; Ejaz, M.; Samy, M. M.; Mousa, A. O.; Kuo, S. W. Design and synthesis of bisulfone-linked two-dimensional conjugated microporous polymers for CO<sub>2</sub> adsorption and energy storage. *Molecules* **2023**, *28*, 3234.
- (33) Mousa, A. O.; Chuang, C. H.; Kuo, S. W.; Mohamed, M. G. Strategic Design and Synthesis of Ferrocene Linked Porous Organic Frameworks toward Tunable CO<sub>2</sub> Capture and Energy Storage. *Int. J. Mol. Sci.* **2023**, *24*, 12371.
- (34) Mahmood, N.; Zhang, C.; Yin, H.; Hou, Y. Graphene-based nanocomposites for energy storage and conversion in lithium batteries, supercapacitors and fuel cells. *J. Mater. Chem. A* **2014**, *2*, 15–32.
- (35) Mohamed, M. G.; Chang, W. C.; Chaganti, S. V.; Sharma, S. U.; Lee, J. T.; Kuo, S. W. Dispersion of ultrastable crown-ether-functionalized triphenylamine and pyrene-linked porous organic conjugated polymers with single-walled carbon nanotubes as high-performance electrodes for supercapacitors. *Polym. Chem.* **2023**, *14*, 589–4601.
- (36) Weng, T. H.; Mohamed, M. G.; Sharma, S. U.; Chaganti, S. V.; Samy, M. M.; Lee, J. T.; Kuo, S. W. Ultrastable three-dimensional triptycene-and tetraphenylethene-conjugated microporous polymers for energy storage. *ACS Appl. Energy Mater.* **2022**, *5*, 14239–14249.
- (37) Ejaz, M.; Mohamed, M. G.; Chang, W. C.; Kuo, S. W. Synthesis and design of hypercrosslinked porous organic frameworks containing tetraphenylpyrazine unit for high-performance supercapacitor. *J. Polym. Sci.* **2024**, *62*, 1629–1638.

- (38) Borenstein, A.; Hanna, O.; Attias, R.; Luski, S.; Brousse, T.; Aurbach, D. Carbon-based composite materials for supercapacitor electrodes: a review. *J. Mater. Chem. A* **2017**, *5*, 12653–12672.
- (39) Simon, P.; Gogotsi, Y. Materials for electrochemical capacitors. *Nat. Mater.* **2008**, *7*, 845–854.
- (40) Samy, M. M.; Mohamed, M. G.; Sharma, S. U.; Chaganti, S. V.; Lee, J. T.; Kuo, S. W. An Ultrastable Tetrabenzonaphthalene-Linked conjugated microporous polymer functioning as a high-performance electrode for supercapacitors. *J. Taiwan Inst. Chem. Eng.* **2024**, *158*, No. 104750.
- (41) Wang, Y. G.; Li, H. Q.; Xia, Y. Y. Ordered whiskerlike polyaniline grown on the surface of mesoporous carbon and its electrochemical capacitance performance. *Adv. Mater.* **2006**, *18*, 2619–2623.
- (42) Zhou, Y.; Qin, Z. Y.; Li, L.; Zhang, Y.; Wei, Y. L.; Wang, L. F.; Zhu, M. F. Polyaniline/multi-walled carbon nanotube composites with core-shell structures as supercapacitor electrode materials. *Electrochim. Acta* **2010**, *55*, 3904–3908.
- (43) Mousa, A. O.; Lin, Z. I.; Chaganti, S. V.; Chuang, C. H.; Chen, C. K.; Kuo, S. W.; Mohamed, M. G. Bifunctional imidazolium linked tetraphenylethene based conjugated microporous polymers for dynamic antibacterial properties and supercapacitor electrodes. *Polym. Chem.* **2024**, *15*, 397–411.
- (44) Lee, J. S. M.; Cooper, A. I. Advances in conjugated microporous polymers. *Chem. Rev.* **2020**, *120*, 2171–2214.
- (45) Liao, Y.; Wang, H.; Zhu, M.; Thomas, A. Efficient supercapacitor energy storage using conjugated microporous polymer networks synthesized from Buchwald–Hartwig coupling. *Adv. Mater.* **2018**, *30*, No. 1705710.
- (46) Mousa, A. O.; Lin, Z. I.; Chuang, C. H.; Chen, C. K.; Kuo, S. W.; Mohamed, M. G. Rational design of bifunctional microporous organic polymers containing anthracene and triphenylamine units for energy storage and biological applications. *Int. J. Mol. Sci.* **2023**, *24*, 8966.
- (47) Mohamed, M. G.; Elsayed, M. H.; Li, C. J.; Hassan, A. E.; Mekhemer, I. M. A.; Musa, A. F.; Hussien, M. K.; Chen, L. C.; Chen, K. H.; Chou, H. H.; Kuo, S. W. Reticular design and alkyne bridge engineering in donor- $\pi$ -acceptor type conjugated microporous polymers for boosting photocatalytic hydrogen evolution. *J. Mater. Chem. A* **2024**, *12*, 7693–7710.
- (48) Mohamed, M. G.; EL-Mahdy, A. F. M.; Kotp, M. G.; Kuo, S. W. Advances in porous organic polymers: Syntheses, structures, and diverse applications. *Mater. Adv.* **2022**, *3*, 707–733.
- (49) Zhang, W.; Zuo, H.; Cheng, Z.; Shi, Y.; Guo, Z.; Meng, N.; Thomas, A.; Liao, Y. Macroscale conjugated microporous polymers: Controlling versatile functionalities over several dimensions. *Adv. Mater.* **2022**, *34*, No. 2104952.
- (50) Chen, L.; Chen, B.; Kang, J.; Yan, Z.; Jin, Y.; Yan, H.; Chen, S.; Xia, C. The synthesis of a novel conjugated microporous polymer and application on photocatalytic removal of uranium (VI) from wastewater under visible light. *Chem. Eng. J.* **2022**, *431*, No. 133222.
- (51) Singh, P. N.; Mohamed, M. G.; Chaganti, S. V.; Sharma, S. U.; Ejaz, M.; Lee, J. T.; Kuo, S. W. Rational design of ultrastable conjugated microporous polymers based on pyrene and perylene units as high-performance organic electrode materials for supercapacitor applications. *ACS Appl. Energy Mater.* **2023**, *6*, 8277–8287.
- (52) Mousa, A. O.; Mohamed, M. G.; Lin, Z. I.; Chuang, C. H.; Chen, C. K.; Kuo, S. W. Conjugated microporous polymers as a novel generation of drug carriers: A systemic study toward efficient carriers of tetracycline antibiotic. *Eur. Polym. J.* **2023**, *196*, No. 112254.
- (53) Mousa, A. O.; Mohamed, M. G.; Lin, Z. I.; Chuang, C. H.; Chen, C. K.; Kuo, S. W. Construction of cationic conjugated microporous polymers containing pyrene units through post-cationic modification for enhanced antibacterial performance. *J. Taiwan Inst. Chem. Eng.* **2024**, *157*, No. 105448.
- (54) Said, A. I.; Mohamed, M. G.; Madhu, M.; Singh, P. N.; Chaganti, S. V.; Elsayed, M. H.; Tseng, W. L.; Raymo, F. M.; Kuo, S. W. Bifunctional luminescent conjugated microporous polymers containing BODIPY and tetraphenylethene units for highly efficient energy storage and enhanced sensing of Cu<sup>2+</sup> ions. *Polymer* **2024**, *300*, No. 126988.
- (55) Sharma, S. U.; Elsayed, M. H.; Mekhemer, I. M. A.; Meng, T. S.; Chou, H. H.; Kuo, S. W.; Mohamed, M. G. Rational design of pyrene and thienyltriazine-based conjugated microporous polymers for high-performance energy storage and visible-light photocatalytic hydrogen evolution from water. *Giant* **2024**, *17*, No. 100217.
- (56) Mohamed, M. G.; Hu, H. Y.; Santhoshkumar, S.; Madhu, M.; Mansoure, T. H.; Hsiao, C. W.; Ye, Y.; Huang, C. W.; Tseng, W. L.; Kuo, S. W. Design and Synthesis of Bifunctional Conjugated Microporous Polymers Containing Tetraphenylethene and Bisulfone Units for Energy Storage and Fluorescent Sensing of p-Nitrophenol. *Colloids Surf. A Physicochem. Eng. Asp.* **2024**, *680*, No. 132675.
- (57) Chang, S. Y.; Elewa, A. M.; Mohamed, M. G.; Mekhemer, I. M. A.; Samy, M. M.; Zhang, K.; Chou, H. H.; Kuo, S. W. Rational design and synthesis of bifunctional Dibenzo [g,p] chrysene-based conjugated microporous polymers for energy storage and visible light-driven photocatalytic hydrogen evolution. *Mater. Today Chem.* **2023**, *33*, No. 101680.
- (58) Li, X. C.; Zhang, Y.; Wang, C. Y.; Wan, Y.; Lai, W. Y.; Pang, H.; Huang, W. Redox-active triazatruxene-based conjugated microporous polymers for high-performance supercapacitors. *Chem. Sci.* **2017**, *8*, 2959–2965.
- (59) Luo, B.; Chen, Y.; Zhang, Y.; Huo, J. Nitrogen-rich anthraquinone–triazine conjugated microporous polymer networks as high-performance supercapacitor. *New J. Chem.* **2021**, *45*, 17278–17286.
- (60) Mohamed, M. G.; Chaganti, S. V.; Sharma, S. U.; Samy, M. M.; Ejaz, M.; Lee, J. T.; Zhang, K.; Kuo, S. W. Constructing conjugated microporous polymers containing the pyrene-4, 5, 9, 10-tetraone unit for energy storage. *ACS Appl. Energy Mater.* **2022**, *5*, 10130–10140.
- (61) Che, H.; Wang, J.; Wang, P.; Ao, Y.; Chen, J.; Gao, X.; Zhu, F.; Liu, B. Simultaneously Achieving Fast Intramolecular Charge Transfer and Mass Transport in Holey D- $\pi$ -A Organic Conjugated Polymers for Highly Efficient Photocatalytic Pollutant Degradation. *JACS Au* **2023**, *3*, 1424–1434.
- (62) Weng, W.; Lin, Z.; Zhang, H.; Niu, F.; Wang, C.; Hu, K.; Guo, J. Effect of ES IPT-Induced Photoisomerization of Keto–Enamine Linkages on the Photocatalytic Hydrogen Evolution Performance of Covalent Organic Frameworks. *JACS Au* **2023**, *3*, 3391–3399.
- (63) Teng, L.; Duan, J.; Liu, H.; Zhang, X.; Li, J.; Li, Y.; Hong, J.; Lyu, W.; Liao, Y. A conjugated microporous polymer–graphene composite porous sandwich-like film for highly efficient flexible supercapacitors. *J. Mater. Chem. A* **2024**, *12*, 12423–12434.
- (64) Mohamed, M. G.; Zhang, X.; Mansoure, T. H.; EL-Mahdy, A. F. M.; Huang, C. F.; Danko, M.; Xin, Z.; Kuo, S. W. Hypercrosslinked porous organic polymers based on tetraphenylanthraquinone for CO<sub>2</sub> uptake and high-performance supercapacitor. *Polymer* **2020**, *205*, No. 122857.
- (65) Liu, J.; Zhou, Y.; Xing, G.; Qi, M.; Tang, Z.; Terasaki, O.; Chen, L. 2D Conductive Metal–Organic Framework with Anthraquinone Built-In Active Sites as Cathode for Aqueous Zinc Ion Battery. *Adv. Funct. Mater.* **2024**, *34*, No. 2312636.
- (66) Ruidas, S.; Pradhan, L.; Mohanty, B.; Dalapati, S.; Kumar, S.; Jena, B. K.; Bhaumik, A. Imine-Linked  $\pi$ -Conjugated Covalent Organic Frameworks as an Efficient Electrode Material for Pseudocapacitive Energy Storage. *ACS Appl. Energy Mater.* **2024**, *7*, 2872–2880.
- (67) Kumar, Y.; Ahmad, I.; Rawat, A.; Pandey, R. K.; Mohanty, P.; Pandey, R. Flexible Linker-Based Triazine-Functionalized 2D Covalent Organic Frameworks for Supercapacitor and Gas Sorption Applications. *ACS Appl. Mater. Interfaces* **2024**, *16*, 11605–11616.
- (68) Tian, L.; Zhi, Y.; Yu, Q.; Xu, Q.; Demir, M.; Colak, S. G.; Farghaly, A. A.; Wang, L.; Hu, X. Enhanced CO<sub>2</sub> Adsorption Capacity in Highly Porous Carbon Materials Derived from Melamine-Formaldehyde Resin. *Energy Fuels* **2024**, *38*, 13186–13195.
- (69) Liu, C.; Zhi, Y.; Yu, Q.; Tian, L.; Demir, M.; Colak, S. G.; Farghaly, A. A.; Wang, L.; Hu, X. Sulfur-Enriched Nanoporous

Carbon: A Novel Approach to CO<sub>2</sub> Adsorption. *ACS Appl. Nano Mater.* **2024**, *7*, 5434–5441.

(70) Lu, T.; Bai, J.; Huang, J.; Yu, Q.; Demir, M.; Kilic, M.; Altay, B. N.; Wang, L.; Hu, X. Self-Activating Approach for Synthesis of 2,6-Naphthalene Disulfonate Acid Disodium Salt-Derived Porous Carbon and CO<sub>2</sub> Capture Performance. *Energy Fuels* **2023**, *37*, 3886–3893.

(71) Yan, Z.; Ren, H.; Ma, H.; Yuan, R.; Yuan, Y.; Zou, X.; Sun, F.; Zhu, G. Construction and sorption properties of pyrene-based porous aromatic frameworks. *Microporous Mesoporous Mater.* **2013**, *173*, 92–98.

(72) Dawson, R.; Adams, D. J.; Cooper, A. I. Chemical tuning of CO<sub>2</sub> sorption in robust nanoporous organic polymers. *Chem. Sci.* **2011**, *2*, 1173–1177.

(73) Yuan, R.; Ren, H.; Yan, Z.; Wang, A.; Zhu, G. Robust tri(4-ethynylphenyl)amine-based porous aromatic frameworks for carbon dioxide capture. *Polym. Chem.* **2014**, *5*, 2266–2272.

(74) Ratvijitvech, T.; Dawson, R.; Laybourn, A.; Khimyak, Y. Z.; Adams, D. J.; Cooper, A. I. Post-synthetic modification of conjugated microporous polymers. *Polymer* **2014**, *55*, 321–325.

(75) Mohamed, M. G.; Sharma, S. U.; Wang, P. T.; Ibrahim, M.; Lin, M. H.; Liu, C. L.; Ejaz, M.; Yen, H. J.; Kuo, S. W. Construction of Fully  $\pi$ -Conjugated, Diyne-Linked Conjugated Microporous Polymers Based on Tetraphenylethene and Dibenzo[*g,p*]chrysene Units for Energy Storage. *Polym. Chem.* **2024**, *15*, 2827–2839.

(76) Mohamed, M. G.; Ibrahim, M.; Chen, N. P.; Basit, A.; Kao, Y. C.; Mousa, A. O.; Samy, M. M.; Kuo, S. W. Tetrabenzonaphthalene and Redox-Active Anthraquinone-Linked Conjugated Microporous Polymers as Organic Electrodes for Enhanced Energy Storage Efficiency. *ACS Appl. Energy Mater.* **2024**, *7*, 5582–5593.

(77) Zhao, X.; Sajjad, M.; Zheng, Y.; Zhao, M.; Li, Z.; Wu, Z.; Kang, K.; Qiu, L. Covalent organic framework templated ordered nanoporous C<sub>60</sub> as stable energy efficient supercapacitor electrode material. *Carbon* **2021**, *182*, 144–154.